



Slow Crack Growth of Germanium

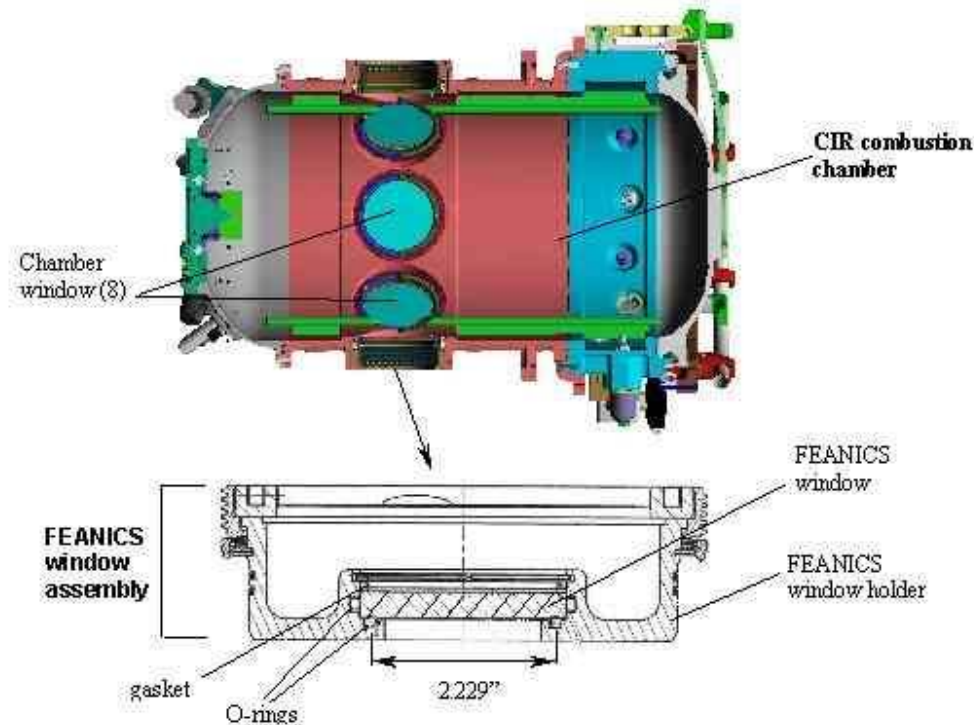
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NASA GRC

ICACC, January 2016



Germanium

- Good electromagnetic transmission in 2-15 μm range. Specialty window material:





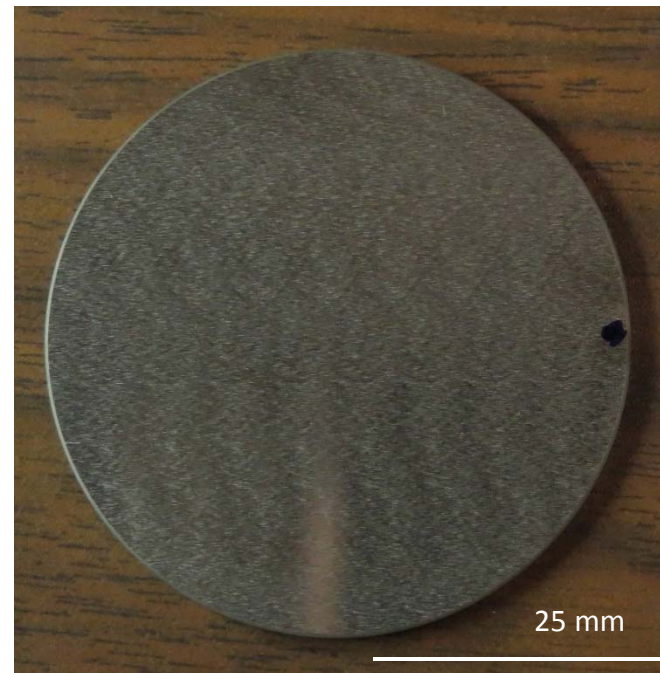
Germanium

- Brittle transition metal.
- Relatively soft.
- Behaves like a soft, brittle ceramic.
- Stress corrosion cracking?
- What is the fracture toughness?



Material

- Single crystal beams
- Coarse grained disks (2" & 5" ϕ):

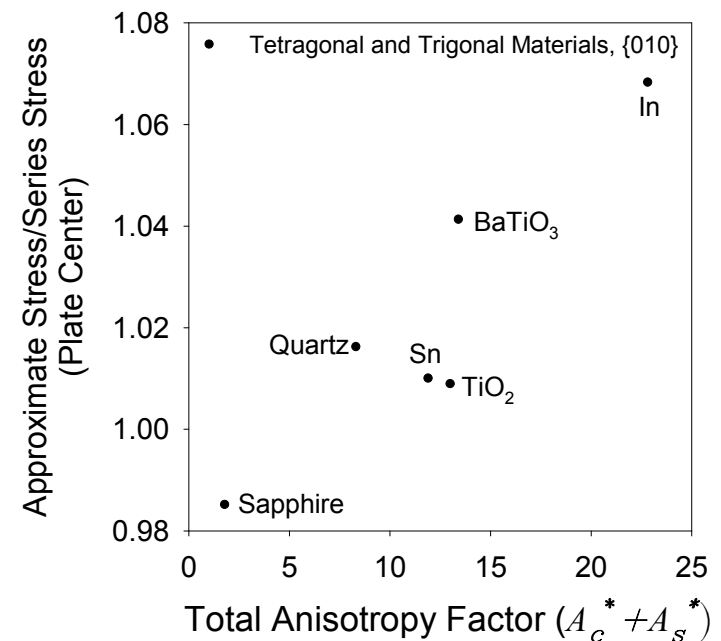
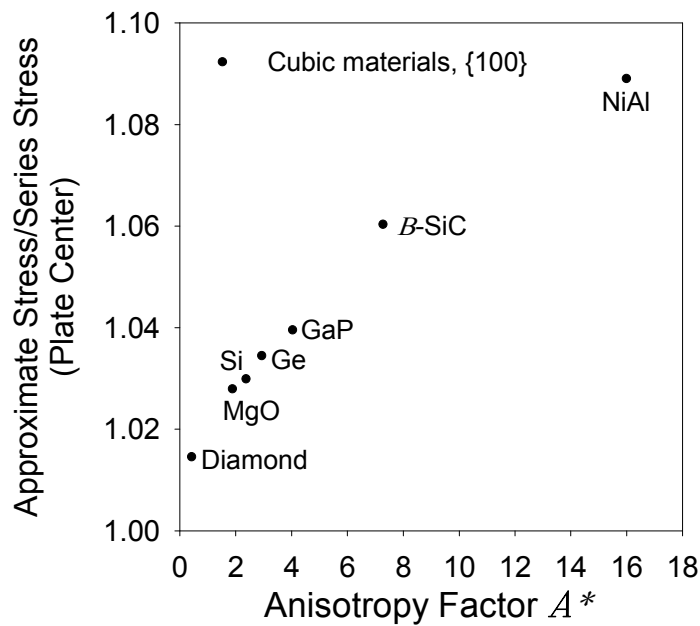


- Variable grain structure – not ideal for testing.



Anisotropy

- Anisotropy factor A^* measures relative magnitude of elastic anisotropy exhibited by a crystal. $A^* = 0$ for isotropic materials, $A^* = 0$ to 1 for many single crystals.



- Relatively low. Running mechanical test on off-axis planes is problematic if the anisotropy is large.



Young's Modulus

- impulse excitation -

- $E_{<111>} = 154.8 \pm 0.9$ GPa
 - $E_{<110>} = 138.3 \pm 0.2$
 - $E_{<100>} = 103.1 \pm 0.6$
- }
- $E_{poly} = 131, \nu_{poly} = 0.21$

| Aggregate Constants | | |
|---------------------|-----|-------|
| GPa | E | ν |
| Voigt | 135 | 0.20 |
| Hashin | 133 | 0.21 |
| Shtrikman | 132 | 0.21 |
| Reuss | 129 | 0.21 |

| Ge | McSkimin | Bogardus | McSkimin | Mason | Average | NASA | % Diff. |
|------------------------------|----------|----------|----------|-------|---------|-------|---------|
| <i>Young's Modulus (GPa)</i> | | | | | | | |
| $E_{<100>} =$ | 104.4 | 102.0 | 102.2 | 103.7 | 103.1 | 103.1 | 0.0% |
| $E_{<110>} =$ | 138.7 | 136.7 | 137.0 | 138.0 | 137.6 | 138.3 | 0.5% |
| $E_{<111>} =$ | 155.8 | 154.2 | 154.5 | 155.1 | 154.9 | 154.8 | -0.1% |

- Well oriented germanium....

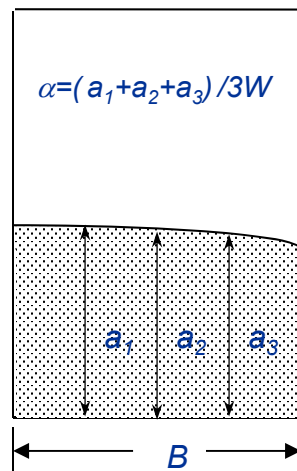


Procedure

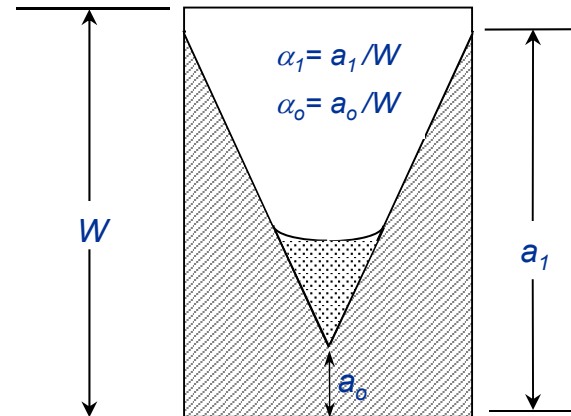
- Fracture Toughness -

- Three standard test methods (C1421):

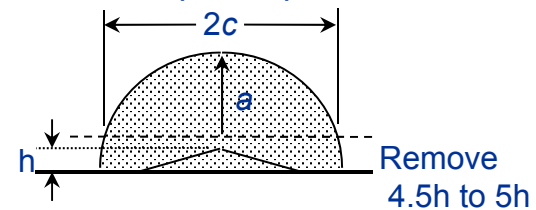
Precracked Beam
(SEPB)



Chevron Notch Beam
(CNB)

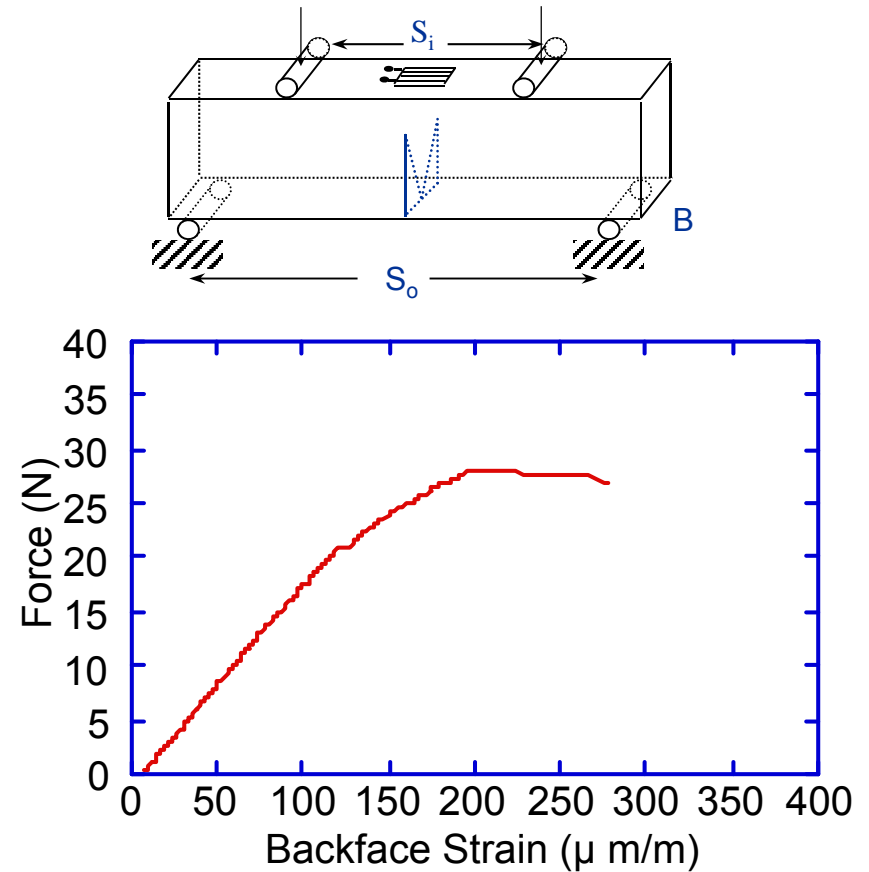
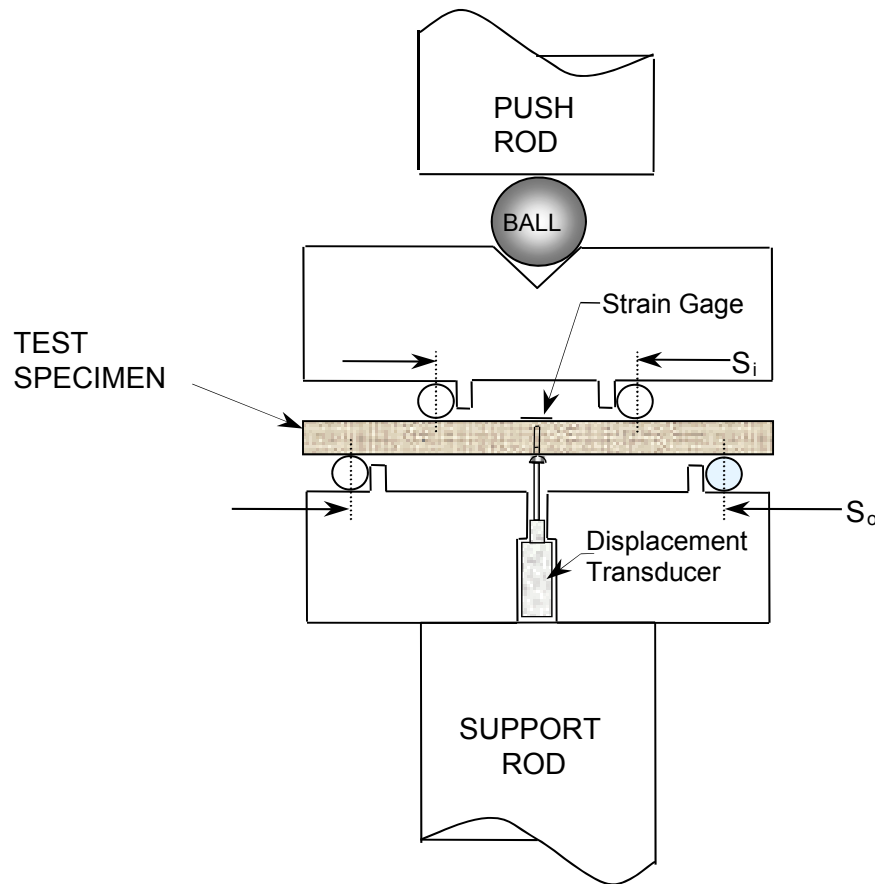


Surface Crack Flexure
(SCF)



- Different crack size
- Different crack formation history
- Different effort

Loading Configuration - Fracture Toughness -



- Relatively simple fixtures: test frame, load cell, recording device.



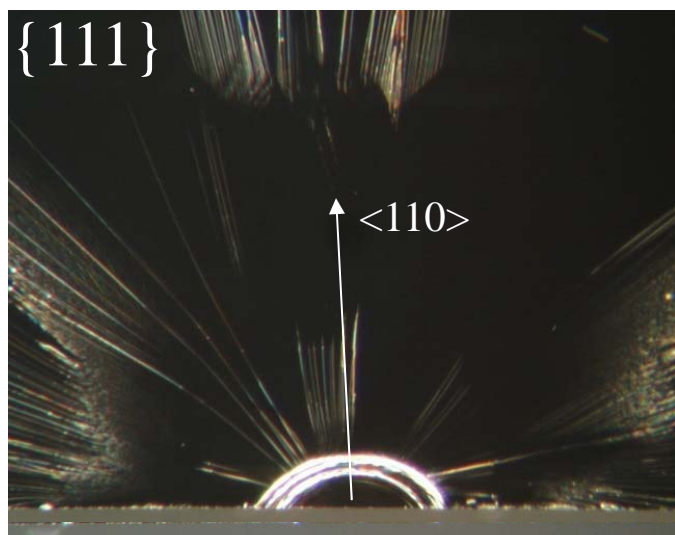
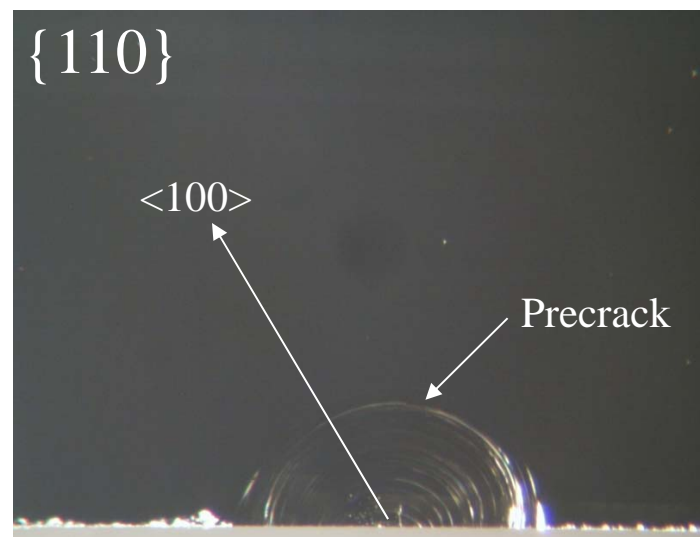
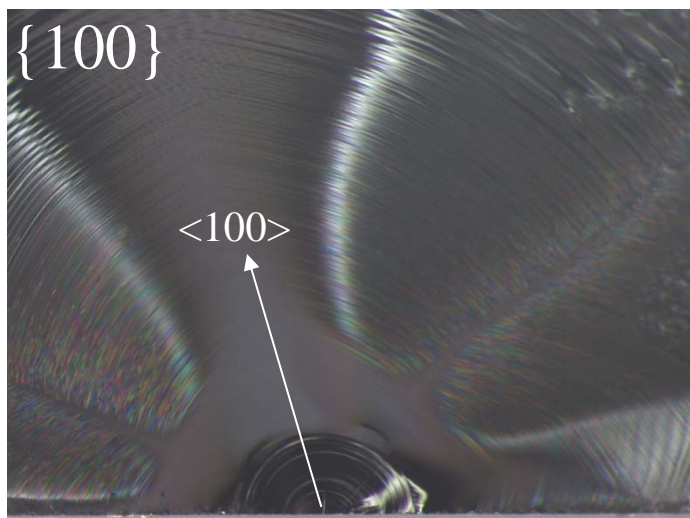
Fracture Toughness

| Method | {100} | {110} | {111} |
|--------|-----------------|-----------------|-----------------|
| SEPB | 0.68 ± 0.04 | 0.68 ± 0.01 | < 0.74 |
| SCF | 0.74 ± 0.02 | 0.74 ± 0.02 | 0.74 ± 0.02 |
| CNB | In progress | In progress | In progress |

- Essentially the same on all planes
- $K_{Iscf\{jkl\}} = 0.74 \pm 0.02 \text{ MPa}\sqrt{\text{m}}$
- $K_{Ipb\{100, 110\}} = 0.68 \pm 0.04 \text{ MPa}\sqrt{\text{m}}$
- ~10% difference between SCF and SEPB. Plasticity?
- Practical value of $K_{I\{jkl\}} = 0.70 \pm 0.02 \text{ MPa}\sqrt{\text{m}}$.



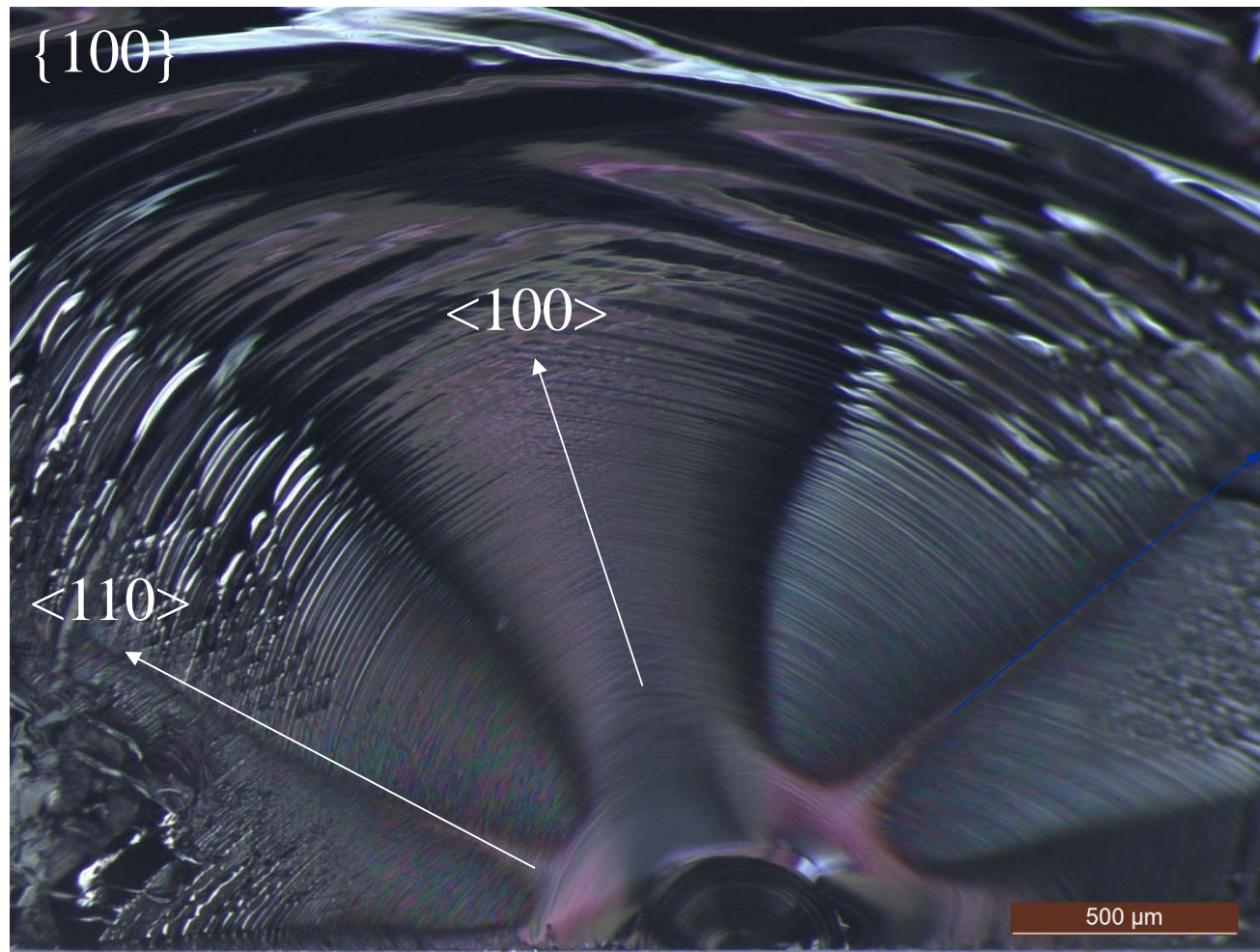
SCF Precracks



- » $\{100\}$ exhibit cathedral Wallner lines.
- » The most planar surface occurs on the $\{110\}$.
- » $\{111\}$ tends to exhibit cleavage steps.
- » Secondary orientation was not fixed.



Cathedral Orientation

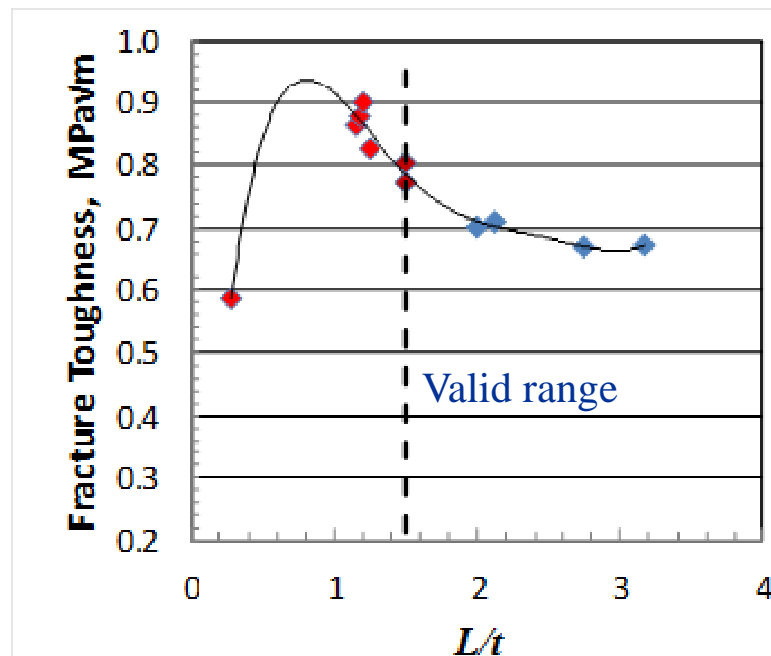


- Peak of cathedral corresponds to the $\langle 100 \rangle \{100\}$.



$K_{I\{111\}}$ Data of Jaccodine

- Reported an energy equivalent value of $0.55 \text{ MPa}\sqrt{\text{m}}$.
- Used DCB w/ fracture mechanics solution that did not include L/t effects.
- Reanalysis gives $K_{I\{111\}} = 0.72 \pm 0.05 \text{ MPa}\sqrt{\text{m}}$ (6) w/ trend toward $0.67 \text{ MPa}\sqrt{\text{m}}$:



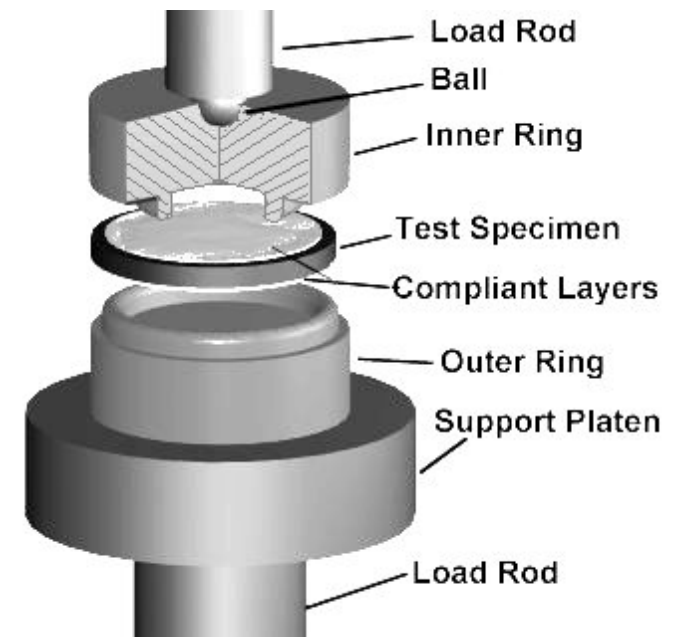
∴ Engineering value
~ $0.68 \text{ MPa}\sqrt{\text{m}}$
for low index planes

R.J. Jaccodine, "Surface Energy of Germanium and Silicon," *J. Electrochemical Soc.*, Vol. 110, No. 6, June, 1963, pp. 524-527.



Strength Testing

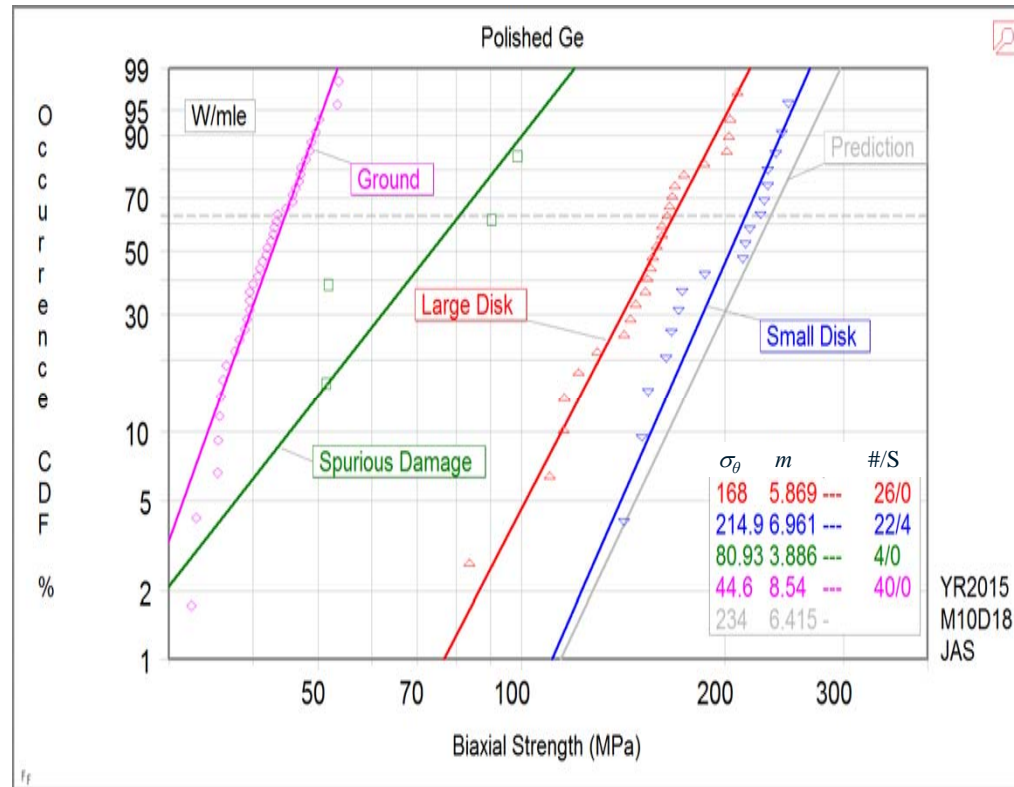
- Constant Stress Rate Tests
(5 MPa/s)
- Biaxial Flexure ring-on-ring (ROR)
- ~400 grit as-ground surfaces in distilled, deionized water
- ~Polished surface in lab air



ASTM C1499



Fracture Strength & Weibull Statistics

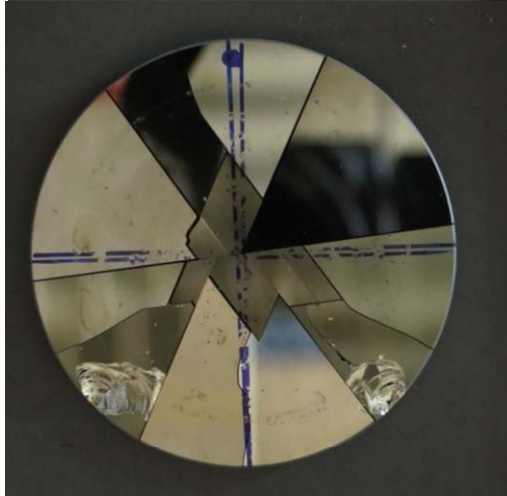
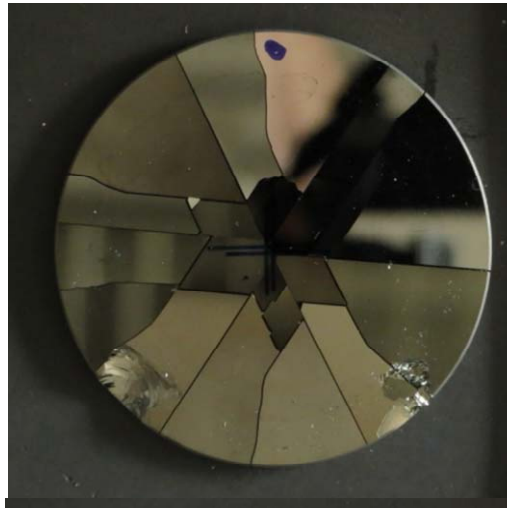
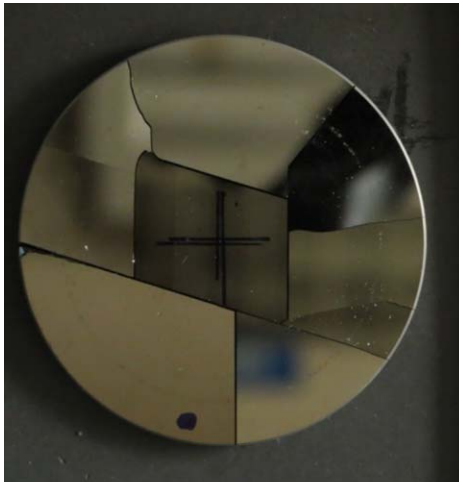


- Polished $m = 6$; ground $m = 9$; spurious damage $m = 4$.
- Scale effect evident: 168 vs 215 MPa.
- Strength of 235 MPa is predicted vs 215 MPa (10%).



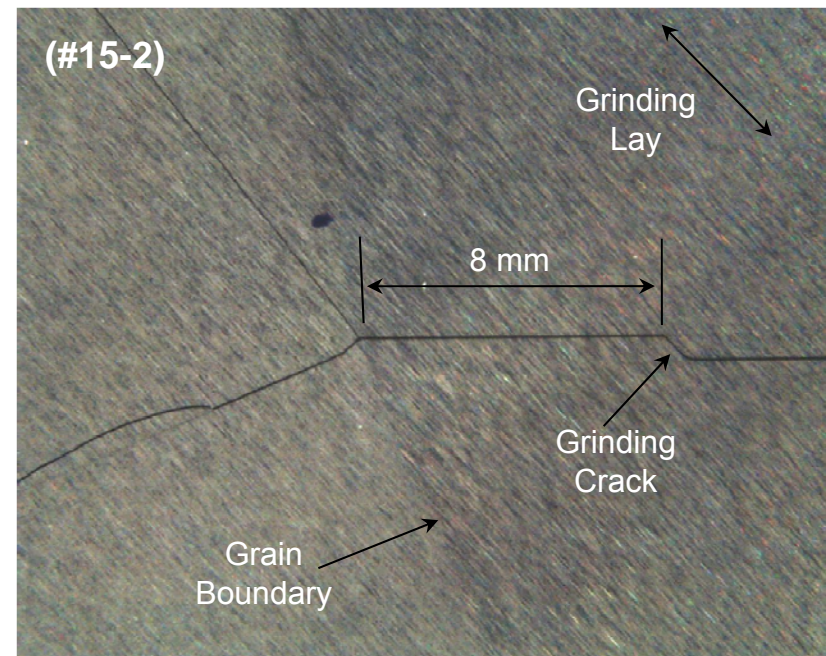
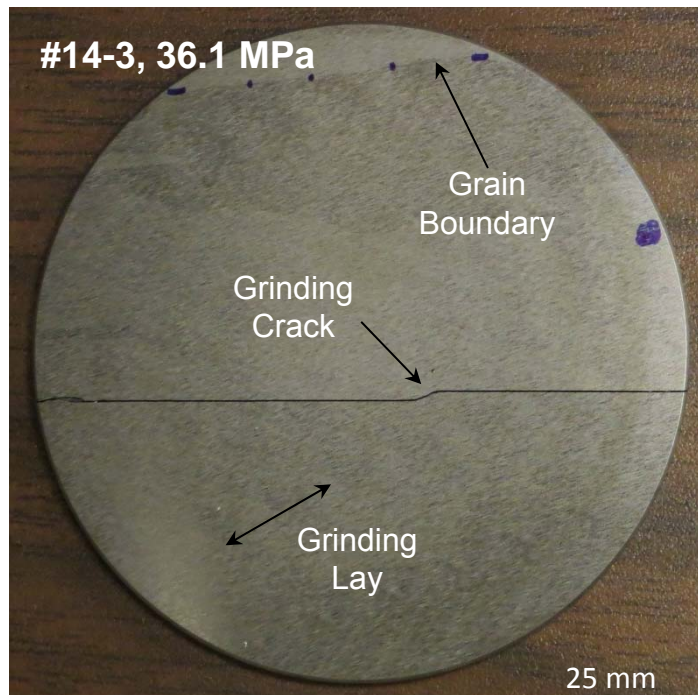
Biaxial Fracture Patterns (polished)

- Repetitive pattern that makes fractography difficult:





Fracture Path - ground disk -

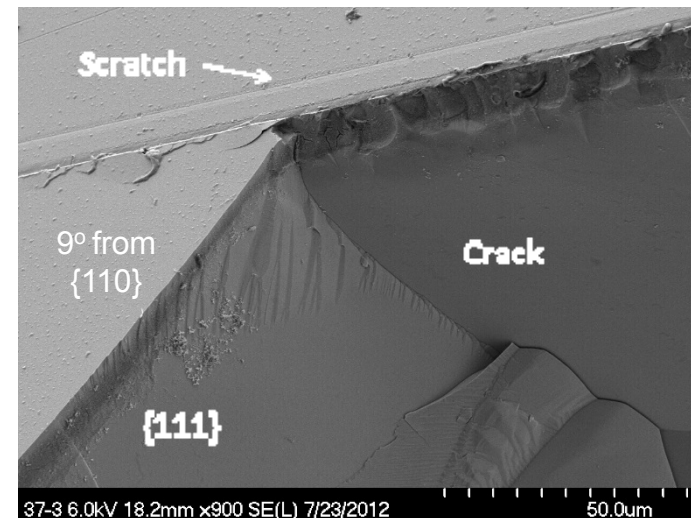
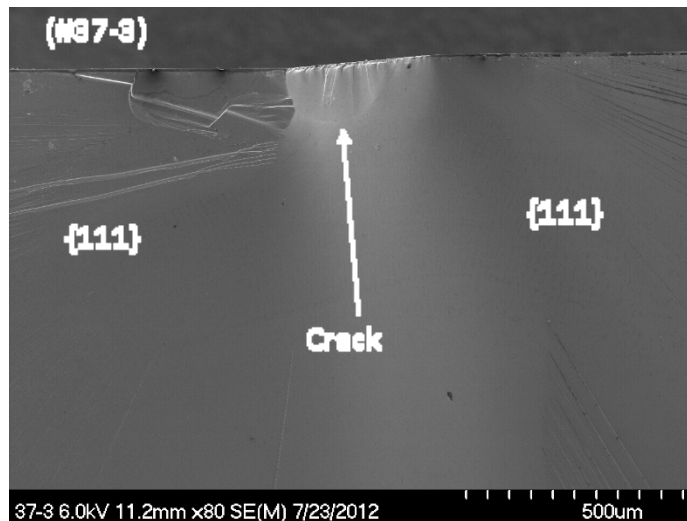


- Crack initiated at a grinding scratch.
- Transited to a low index planes.
- Deflected at a grain boundary.



Fracture Path in a Polished ROR Disk

- Crack initiated from a semi-elliptical crack emanating from a scratch.
- Turned onto the {111} plane:

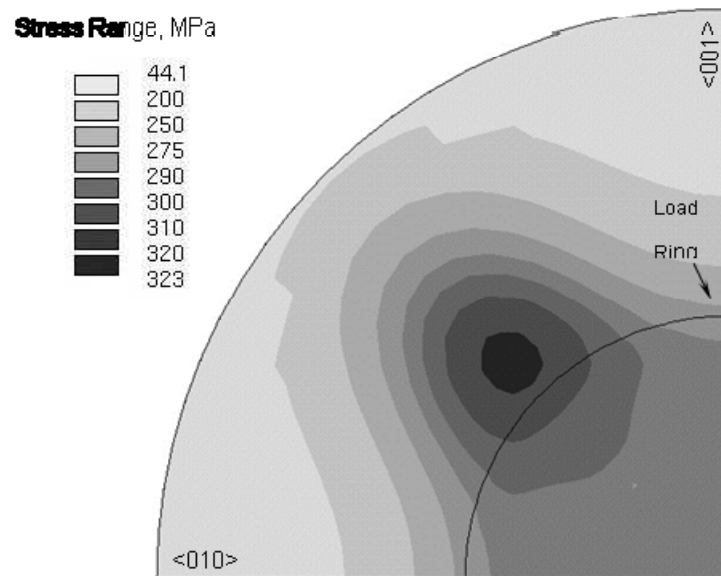


- Opportunity to estimate the fracture toughness!
- $K_{I\{hkl\}} = 0.73 \text{ MPa}\sqrt{\text{m}}$.
- Why did the crack turn?

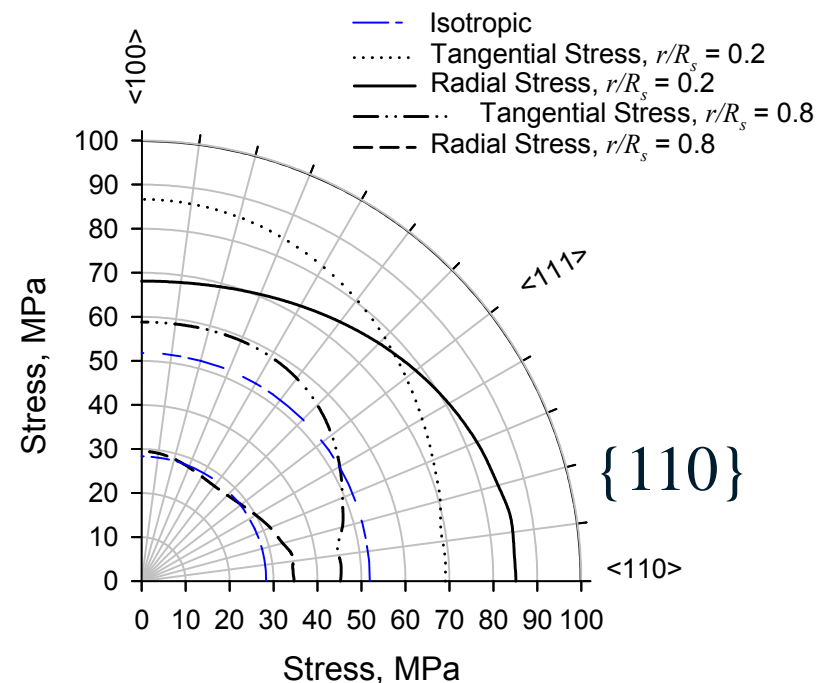


Preferred Fracture Plane

- The fracture toughness on low index planes is similar, so why is the {111} the preferred propagation plane?
- The {111} is the stiffest direction, and stiff directions exhibit high stresses under strain controlled situations.....



Stress concentration where the load ring intersection the stiff direction.

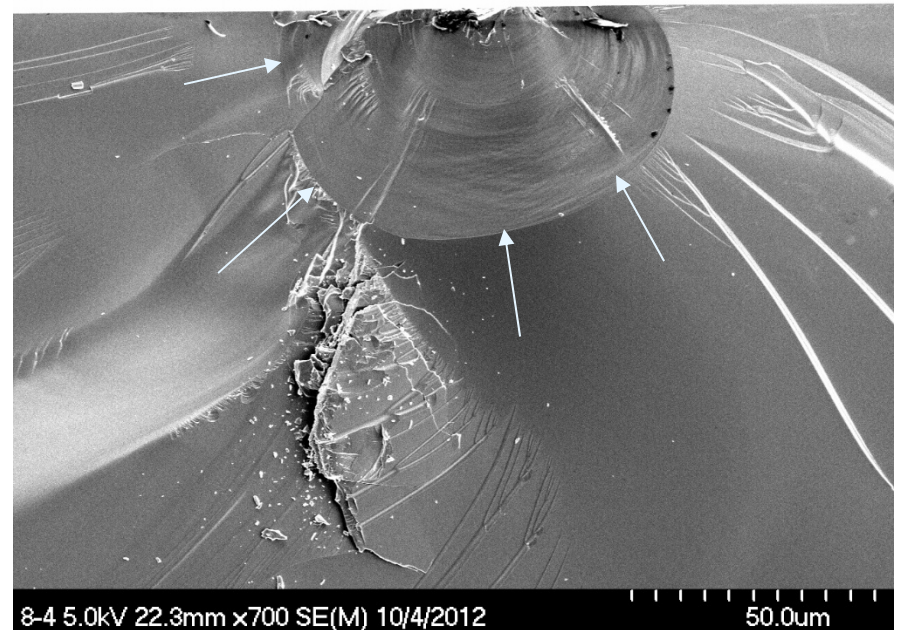
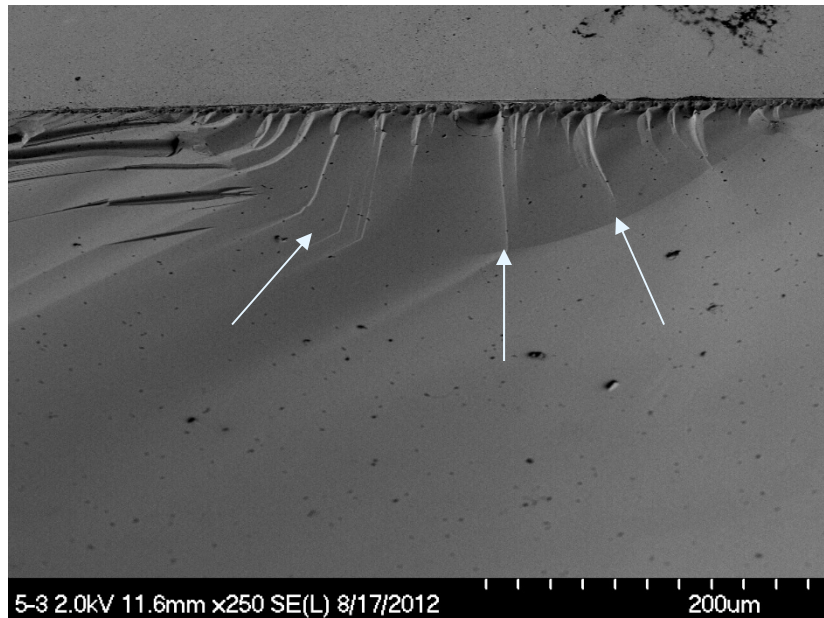


But, for a pressurized plate, the stress concentrations are not exhibited.



Fracture Toughness

– semi-elliptical cracks on high index planes -



- For polished specimens, $K_I = 0.77 \pm 0.04 \text{ MPa}\sqrt{\text{m}}$ (0.73-0.83).
- For grinding cracks, $K_I = 0.87 \pm 0.04 \text{ MPa}\sqrt{\text{m}}$ (0.80 – 0.90).
- Higher due to random orientation and transition to {111}.
- Caveat: local stress not precisely know.....



Slow Crack Growth

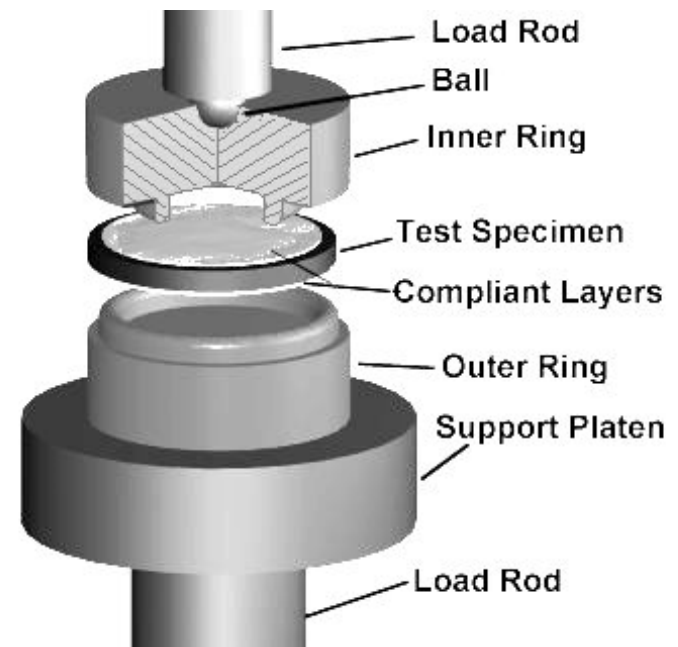
- Experimental Approach -

- Constant Stress Rate Testing “dynamic fatigue”
 - ASTM C1368
- Strength based approach with advantages & disadvantages:
 - rapid test; simple geometry
 - samples the inherent, small flaws
 - statistical scatter (many specimens needed)
 - averaging of fatigue regions



Experimental Procedure

- Constant Stress Rate Tests
(5 to 5×10^{-4} MPa/s)
- Biaxial Flexure (Ring-on-ring)
- Distilled, deionized water
- ~400 grit as-ground surfaces
- ~10 tests per stress rate
- ~40 tests





Slow Crack Growth Analysis

- Crack growth function:

$$v = \frac{da}{dt} = AK_I^n = A * \left[\frac{K_I}{K_{IC}} \right]^n$$

- Constant stress rate testing:

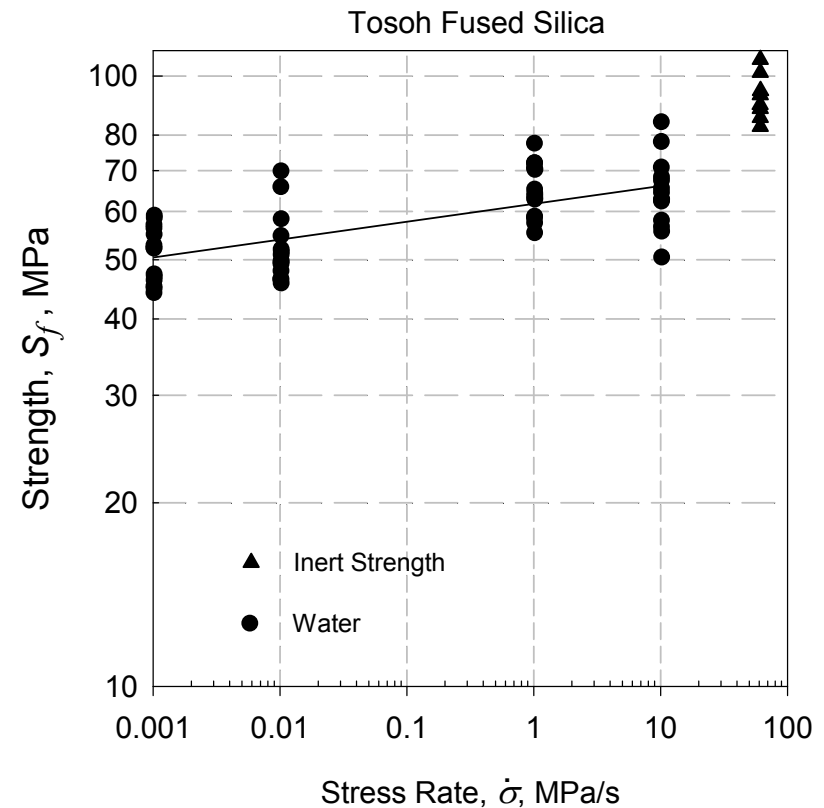
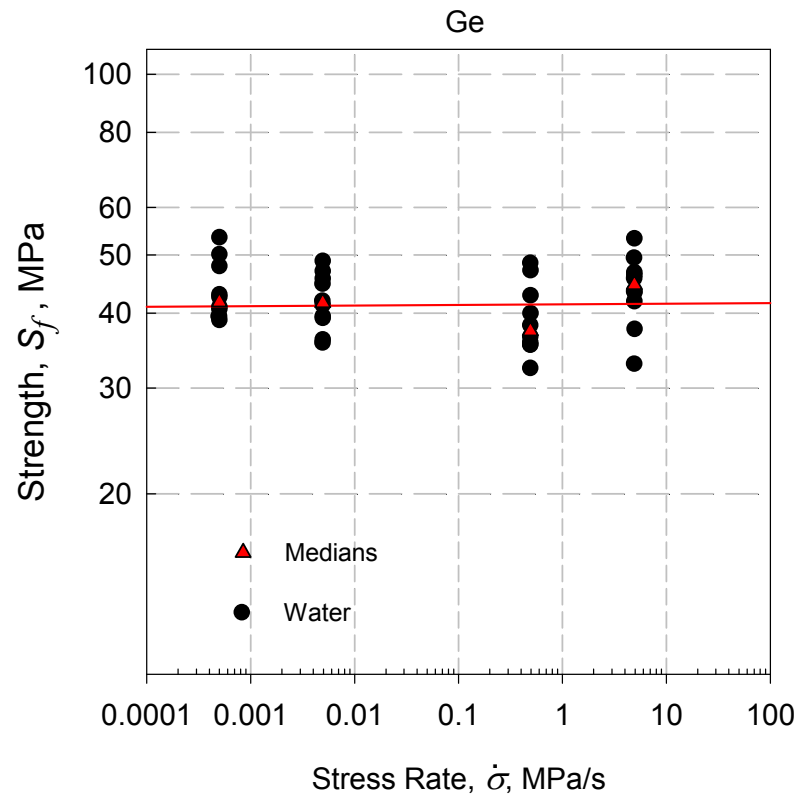
$$\sigma_f = \left[B(n+1)\sigma_i^{n-2}\dot{\sigma} \right]^{1/(n+1)} \quad B = \frac{2K_{Ic}^{2-n}}{AY^2(n-2)} = \frac{2K_{Ic}^2}{A^*Y^2(n-2)}$$

- Parameter extraction via regression:

$$\log_{10} \sigma_f = \underbrace{\frac{1}{n+1} \log_{10} \dot{\sigma}}_{\text{(Slope } \alpha)} + \underbrace{\log_{10} D}_{\text{(Intercept } \beta)} \quad \log_{10} D = \frac{1}{n+1} \log_{10} \left[B(n+1)\sigma_i^{n-2} \right]$$



Constant Stress Rate Curve



- Still some scatter.
- Medians clarify the trend.
- Slope is negative to zero; $n > 100$, no measurable SCG.



Summary and Conclusions

- Ge exhibits similar fracture toughness of $K_I = 0.68 \pm 0.02 \text{ MPa}\sqrt{\text{m}}$ on low index planes. Lower than Si!
- Randomly oriented cracks exhibit higher apparent toughness, but turn and propagate on the stiff $\{111\}$ directions due to higher stresses (??).....FEA.
- Natural cleavage plane appears to be the $\{110\}$.
- Weibull modulus varies from $m = 4$ (spurious) to $m = 9$ (ground).
- Strength varies from $S_f = 40 \text{ MPa}$ (ground) to 160 MPa (polished).
- Ge exhibits a Weibull scale effect, but does not exhibit measurable SCG.



Summary and Conclusions

- Aggregate, polycrystalline Yong's modulus and Poisson's ratio are $E_{poly} = 131 \text{ GPa}$, $\nu_{poly} = 0.21$.
- ROR loading results in stress concentrations at the stiff directions of single crystals.
- From a stress state point-of-view, a lower strength is measurement is expected.....
- However, from an effective area perspective, a high strength should be measured.
- POR maybe a better test method, but more effort.



Acknowledgements

- Thanks to Terry McCue for SEM fractography.
- Thanks to Rick Rogers for x-ray diffraction.
- Thanks to Penni Dalton for funding & reviews.